

# Implanting Polypyrrole in Metal-Porphyrin MOFs: Enhanced Electrocatalytic Performance for CO<sub>2</sub>RR

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Cite This: *ACS Appl. Mater. Interfaces* 2021, 13, 54959–54966



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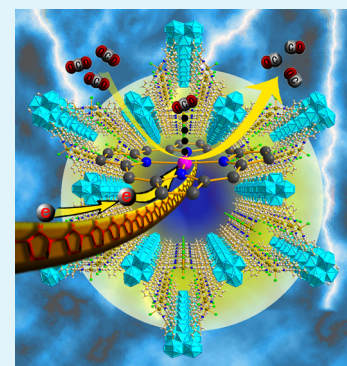


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Supporting Information

**ABSTRACT:** Metal–organic frameworks (MOFs) with plenty of active sites and high porosity have been considered as an excellent platform for the electroreduction of CO<sub>2</sub>, yet they are still restricted by the low conductivity or low efficiency. Herein, we insert the electron-conductive polypyrrole (PPy) molecule into the channel of MOFs through the in situ polymerization of pyrrole in the pore of MOF-545-Co to increase the electron-transfer ability of MOF-545-Co and the obtained hybrid materials present excellent electrocatalytic CO<sub>2</sub>RR performance. For example, FE<sub>CO</sub> of PPy@MOF-545-Co can reach up to 98% at −0.8 V, almost 2 times higher than that of bare MOF-545-Co. The high performance might be attributed to the incorporation of PPy that can serve as electric cables in the channel of MOF to facilitate electron transfer during the CO<sub>2</sub>RR process. This attempt might provide new insights to improve the electrocatalytic performance of MOFs for CO<sub>2</sub>RR.



**KEYWORDS:** polypyrrole, metalloporphyrin organic framework, electrocatalytic performance, electron transfer, CO<sub>2</sub>RR

## INTRODUCTION

The increasing emission amount of carbon dioxide (CO<sub>2</sub>) has led to severe environmental issues and even irreversible climate changes such as sea level rising or land desertification, and so forth.<sup>1</sup> Under this circumstance, converting CO<sub>2</sub> into fuels or other high-valued chemicals is considered as one of the most important ways to address this threat.<sup>2,3</sup> Various techniques for storing or transforming CO<sub>2</sub> have been studied,<sup>4</sup> during which, the electrocatalytic CO<sub>2</sub> reduction reaction (CO<sub>2</sub>RR) as a sort of high-efficiency and controllable method has been deemed as one of the most promising approaches to convert CO<sub>2</sub> into energy products (e.g., CO, HCOOH, C<sub>2</sub>H<sub>5</sub>OH, and CH<sub>4</sub>).<sup>5–10</sup> Over the past years, numerous researchers devote themselves to the development of highly active electrocatalytic materials for CO<sub>2</sub>RR<sup>11–15</sup> and various materials including metals,<sup>16,17</sup> transition-metal chalcogenides,<sup>18</sup> transition-metal oxides,<sup>19</sup> metal-free two-dimensional materials,<sup>20–22</sup> and MOFs<sup>23–28</sup> have been investigated. Several primary principles for the design of CO<sub>2</sub>RR electrocatalysts should be considered: (1) highly active catalytic sites; (2) sufficient surface area; and (3) efficient electron-transfer ability. While many interesting materials have been studied, the low selectivity or instability of the materials has largely hindered the progress of electrocatalytic CO<sub>2</sub>RR to a great extent.

Among various alternatives, the controllable active metal sites and organic ligands of MOFs make it an ideal electrocatalytic platform for electrocatalytic CO<sub>2</sub>RR. The high porosity and plenty of metal sites endow MOFs with favorable capability for establishing desired catalytic environments for CO<sub>2</sub>RR.<sup>29–36</sup> However, the low electron transfer and

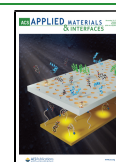
donating ability have largely restricted most of MOFs in the efficient electrocatalytic CO<sub>2</sub>RR, especially for the multiple electron-transfer necessity during the reduction processes. Thus, increasing the charge-transfer ability of catalytic sites in MOFs might enhance the electrocatalytic activity and selectivity for CO<sub>2</sub>RR. To address these issues, various methods have been carried out, such as introducing ligands or metal clusters with high electron density, adding electron-rich molecules into MOF structures,<sup>37–39</sup> and so forth. However, the difficulty in material design or complicated processes have restricted their wide-spread applications and feasible strategies that enable facile processing of MOFs into valuable CO<sub>2</sub>RR electrocatalysts are much desired yet largely unmet.<sup>40–43</sup>

With these considerations in mind, polypyrrole (PPy), as a kind of powerful conductive polymer, comes to our mind as a desired additive to be hybridized with MOFs to achieve efficient CO<sub>2</sub>RR systems. We intend to assemble PPy in the channel of MOF-545-Co for the following reasons: (1) PPy is one of the extensively investigated conductive polymers due to its potential applications in various devices and has been proposed that the introduction of PPy into the structure of

**Received:** August 10, 2021

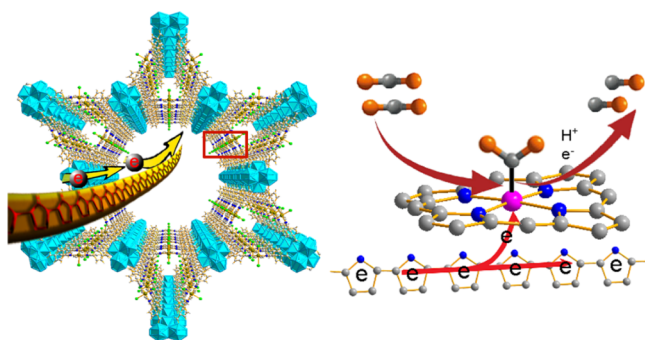
**Accepted:** November 2, 2021

**Published:** November 12, 2021



MOFs will largely improve the electron-transfer ability;<sup>44,45</sup> (2) MOF-545-Co possess high porosity, uniformly distributed active sites, and large pore size that can provide enough space for PPy loading and CO<sub>2</sub>RR; and (3) the synergistic integration of them might create a powerful catalysis system that can endow high electron-transfer ability to MOFs for efficient CO<sub>2</sub>RR. As far as we know, this kind of catalytic system has been rarely reported.

Herein, a feasible in situ low-temperature polymerization method has been reported to synthesize a series of PPy@MOF-545-M (M = Fe, Co, and Ni) hybrid materials and successfully apply them in electrocatalytic CO<sub>2</sub>RR. In the obtained PPy@MOF-545-Co, the PPy molecule is similar to a cable in the MOF channel, which can transfer electrons to the active center of Co-TCPP more smoothly and greatly improve the electrocatalytic activity for CO<sub>2</sub>RR (Figure 1). Specifically,



**Figure 1.** Schematic presentation for the advantages of PPy in the channel of MOF-545-Co for electrocatalytic CO<sub>2</sub>RR.

PPy@MOF-545-Co can convert CO<sub>2</sub> to CO with a high selectivity (98%, −0.8 V), almost 2 times higher than that of MOF-545-Co. Besides, the catalyst can be durable for 10 h with remained crystallinity and property.

## RESULTS AND DISCUSSION

Zr<sub>6</sub>O<sub>8</sub> clusters assembled with Co-TCPP generated two kinds of pores (16 and 36 Å) in the structure of MOF-545-Co, which can supply enough space to encapsulate PPy molecules (diameter about 4.5 Å). Powder X-ray diffraction (PXRD) patterns demonstrate that the framework still maintains the structure of MOF-545-Co after the loading of PPy (Figure 1a). However, the PPy peaks were hardly observed in the PXRD pattern of PPy@MOF-545-Co due to both the small size of PPy molecules and their extremely low amount when compared with the host MOF.<sup>46</sup> The morphology of PPy@MOF-545-Co was investigated by scanning electron microscopy (SEM) and transition electron microscopy (TEM). As shown in Figures 2c,d and Figure S1, the as-prepared PPy@MOF-545-Co displays rod-like shape with a diameter of ~100 nm and a length of ~500 nm. Specifically, the morphology of MOF-545-Co remained unchanged during the loading process of PPy.<sup>37</sup> The PPy-loaded PPy@MOF-545-Co rod shows a smooth surface, which indicates that PPy polymerizes into the pore channel of MOF without the residues on the surface. Moreover, the elements of Co, Zr, C, and N are evenly dispersed in the rod-like particles of PPy@MOF-545-Co as shown in the energy-dispersive X-ray spectroscopy (EDS) elemental mapping images (Figure 2e). To investigate the catalytic performance of PPy in MOF with different metals, the PPy molecule was also introduced into both MOF-545-Fe and

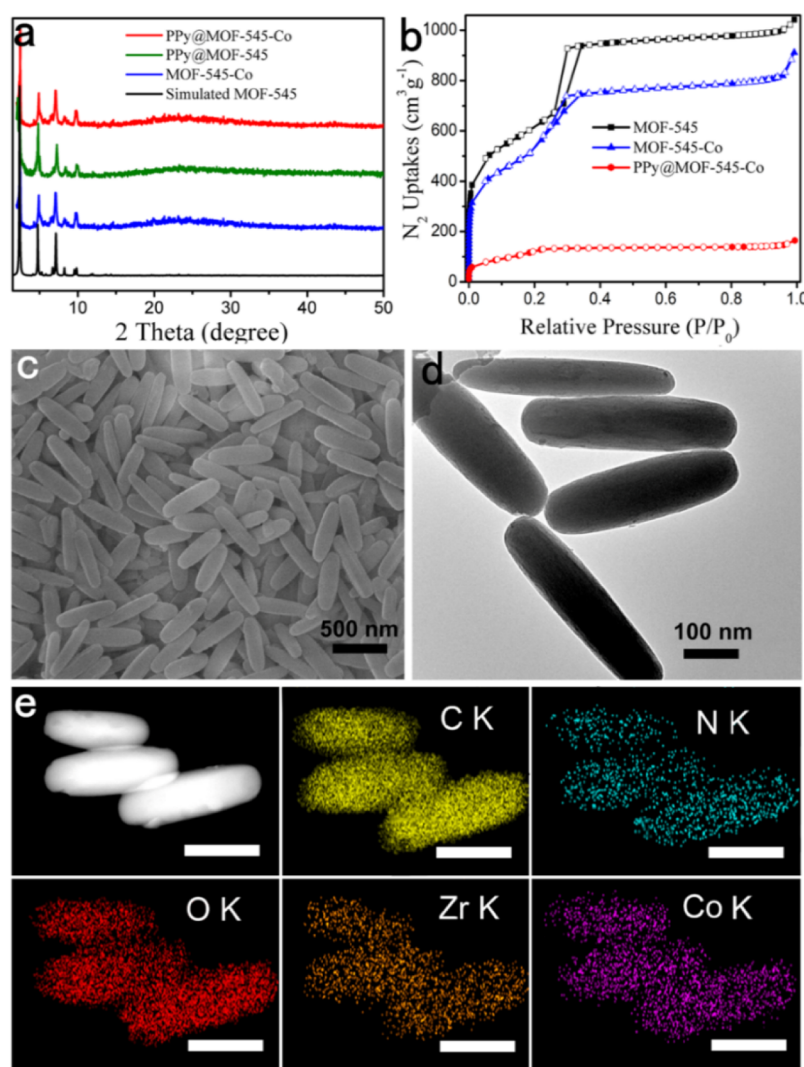
MOF-545-Ni through the similar procedures as that of PPy@MOF-545-Co. The PXRD patterns (Figure S2a) show that PPy@MOF-545-Fe and PPy@MOF-545-Ni can maintain the structure of MOF-545.

The encapsulation of PPy in MOF-545-Co was also proved by the IR spectrum, which displayed the characteristic peaks of PPy (Figure S3). The peaks at 1555 and 964 cm<sup>−1</sup> were ascribed to the stretching vibration of Py rings and PPy, respectively. Besides, the peaks at 1053 and 1306 cm<sup>−1</sup> corresponded to the C–H deformation vibrations and C–N stretching vibrations, respectively.<sup>12,40</sup> In addition, the N<sub>2</sub> sorption isotherms of PPy@MOF-545-Co and the contrast samples (i.e. MOF-545 and MOF-545-Co) were obtained to study the porosity and surface areas of the obtained materials. The Brunauer–Emmett–Teller surface areas (*S*<sub>BET</sub>) of MOF-545, MOF-545-Co, and PPy@MOF-545-Co were calculated to be 1958, 1716, and 146.6 m<sup>2</sup>/g, respectively (Figure 2b and Table S1). This obvious decrease of *S*<sub>BET</sub> and pore volume (Figure S4) resulted from the encapsulation of PPy in the pore of MOF-545-Co. The high-resolution TEM image showed the observed crystalline lattice structure of PPy@MOF-545-Co (Figure S5). When the PPy molecule formed in the MOF channels, the host–guest  $\pi$ – $\pi$  interaction between the PPy molecule and TCPP ligand in MOF might be beneficial for the charge transfer.<sup>46</sup> To further study the effect of PPy on the performance, we introduced different amounts of PPy in MOF-545-Co and tested the catalytic performance for CO<sub>2</sub>RR. The PPy@MOF-545-Co with different PPy loadings was also prepared and characterized by PXRD and N<sub>2</sub> sorption measurement. The PXRD patterns showed that the structure of MOF-545-Co was still maintained after the loading of different amounts of PPy (Figure S6a). The N<sub>2</sub> sorption measurement showed that *S*<sub>BET</sub> increased with the decrease of the PPy loading amount in the MOF channel (Figure S6b and Table S1), which implies that the different loading amount of PPy would partially occupy different pore volume of MOF.

To further support the high porosity, CO<sub>2</sub> adsorption measurements were also performed. From the results of CO<sub>2</sub> uptake curves, the CO<sub>2</sub> capacities of MOF-545, MOF-545-Co, and PPy@MOF-545-Co were tested to be 39, 48, and 12 cm<sup>3</sup>/g at 298 K, respectively (Figure S7a). The difference of CO<sub>2</sub> uptake results from the loading of PPy, which affects the porosity of MOFs. In addition, the adsorption enthalpy of PPy@MOF-545-Co, MOF-545-Co, and MOF-545 (Figure S7b) were calculated based on the CO<sub>2</sub> adsorption curves obtained at 273, 283, and 298 K. The CO<sub>2</sub> adsorption enthalpy of PPy@MOF-545-Co (44 kJ/mol) was higher than that of MOF-545-Co (34.5 kJ/mol) and MOF-545 (19.5 kJ/mol). The results indicated that the introduction of PPy was favorable for the adsorption of CO<sub>2</sub>.

Thermogravimetric analysis (TGA) was also used to test the loading amount of Py in the channels of MOF-545-Co. In the TGA curve of Py@MOF-545-Co, the initial weight loss of 17% from 25 to 200 °C belongs to the loss of Py in the channel, which indicates that the loading of Py is about 17% (Figure S9), which was in accordance with the amount of isolated PPy from PPy@MOF-545-Co.<sup>39,47</sup> This result also indicated that all the adsorbed Py polymerized into PPy in the channel of MOF-545-Co.

The electrocatalytic CO<sub>2</sub>RR tests were conducted in an electrochemical H-type cell. The electrochemical properties of PPy@MOF-545-Co, MOF-545-Co, PPy@MOF-545, and PPy were studied using a three-electrode system in 0.5 M KHCO<sub>3</sub>

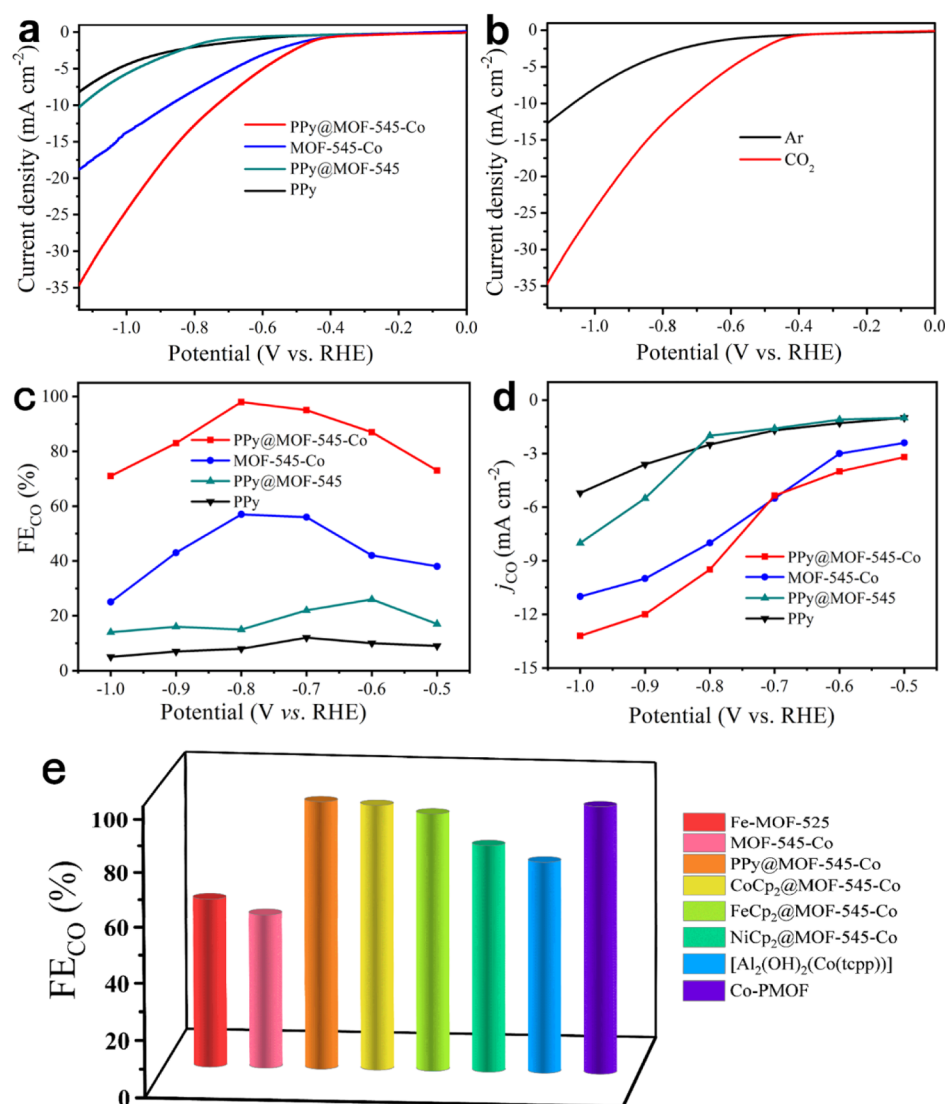


**Figure 2.** Characterization of PPy@MOF-545-Co. (a) PXRD patterns of PPy@MOF-545-Co, MOF-545-Co, MOF-545, and simulated MOF-545. (b)  $N_2$  sorption isotherms of PPy@MOF-545-Co, MOF-545-Co, and MOF-545 measured at 77 K. (c) SEM image of PPy@MOF-545-Co. (d) TEM image of PPy@MOF-545-Co. (e) Scanning TEM-high-angle annular dark field image and EDS elemental mapping of MOF-545-Co (scale bar is 200 nm).

solution. The linear sweep voltammetry (LSV) curves showed the variation trend of current with the increase of voltage. The current of PPy@MOF-545-Co in the LSV curve increased sharply after  $-0.42$  V, and the largest current density was  $32 \text{ mA cm}^{-2}$  at  $-1.1$  V, which was largely higher than that of MOF-545-Co, PPy@MOF-545, and PPy (Figure 3a). The higher current density of PPy@MOF-545-Co indicated the higher electrocatalysis activity for  $\text{CO}_2$  reduction compared with MOF-545-Co, PPy@MOF-545, and PPy. To further detect the electrochemical performance of PPy@MOF-545-Co, the LSV curves were tested in both  $\text{CO}_2$ -saturated and Ar-saturated  $\text{KHCO}_3$  solution (Figure 3b). The higher current density measured under  $\text{CO}_2$ -saturated  $\text{KHCO}_3$  solution than that under Ar-saturated  $\text{KHCO}_3$  solution indicated that the current increasing mainly came from  $\text{CO}_2\text{RR}$  (Figure 3b).<sup>48</sup> The gaseous and liquid products were detected by gas chromatography (GC) and  $^1\text{H}$  nuclear magnetic resonance spectroscopy, respectively (Figure S11). The result exhibited that no liquid product was detected and CO was the main gas product.<sup>49</sup> Moreover, the FEs and current density tests were conducted to study the performances of samples (i.e., PPy@

MOF-545-Co, MOF-545-Co, PPy, and PPy@MOF-545). The  $\text{FE}_{\text{CO}}$  of PPy@MOF-545-Co increased with the increase of potential and reached the peak (98%) at  $-0.8$  V, and the overall  $\text{FE}_{\text{CO}}$  level kept well above 75% under the potentials ranging from  $-0.5$  to  $-1.0$  V [vs reversible hydrogen electrode (RHE)]. The maximum  $\text{FE}_{\text{CO}}$  value (Figure 3c) of PPy@MOF-545-Co was 98% at  $-0.8$  V, which was remarkably higher than that of MOF-545-Co (57%,  $-0.8$  V), PPy (12%,  $-0.7$  V), and PPy@MOF-545 (26%,  $-0.6$  V). The higher CO partial current density of PPy@MOF-545-Co ( $13 \text{ mA cm}^{-2}$  at  $-1.0$  V) than contrast samples indicated the better electron-transfer ability in the  $\text{CO}_2\text{RR}$  process (Figure 3d). Noteworthy, the  $\text{FE}_{\text{CO}}$  value (98%,  $-0.8$  V) was also higher than most of the metalloporphyrin MOFs (Figure 3e).<sup>23,24,33,34</sup> To select the most suitable active metal site in the MOF structure, we selected MOF-545-Co, MOF-545-Ni, and MOF-545-Fe to load PPy and further test their catalytic performances. The result indicated that PPy@MOF-545-Co showed a higher  $\text{FE}_{\text{CO}}$  value than PPy@MOF-545-Fe (89.2%,  $-0.7$  V) and PPy@MOF-545-Ni (87.8%,  $-0.9$  V) (Figure S1b). Therefore,



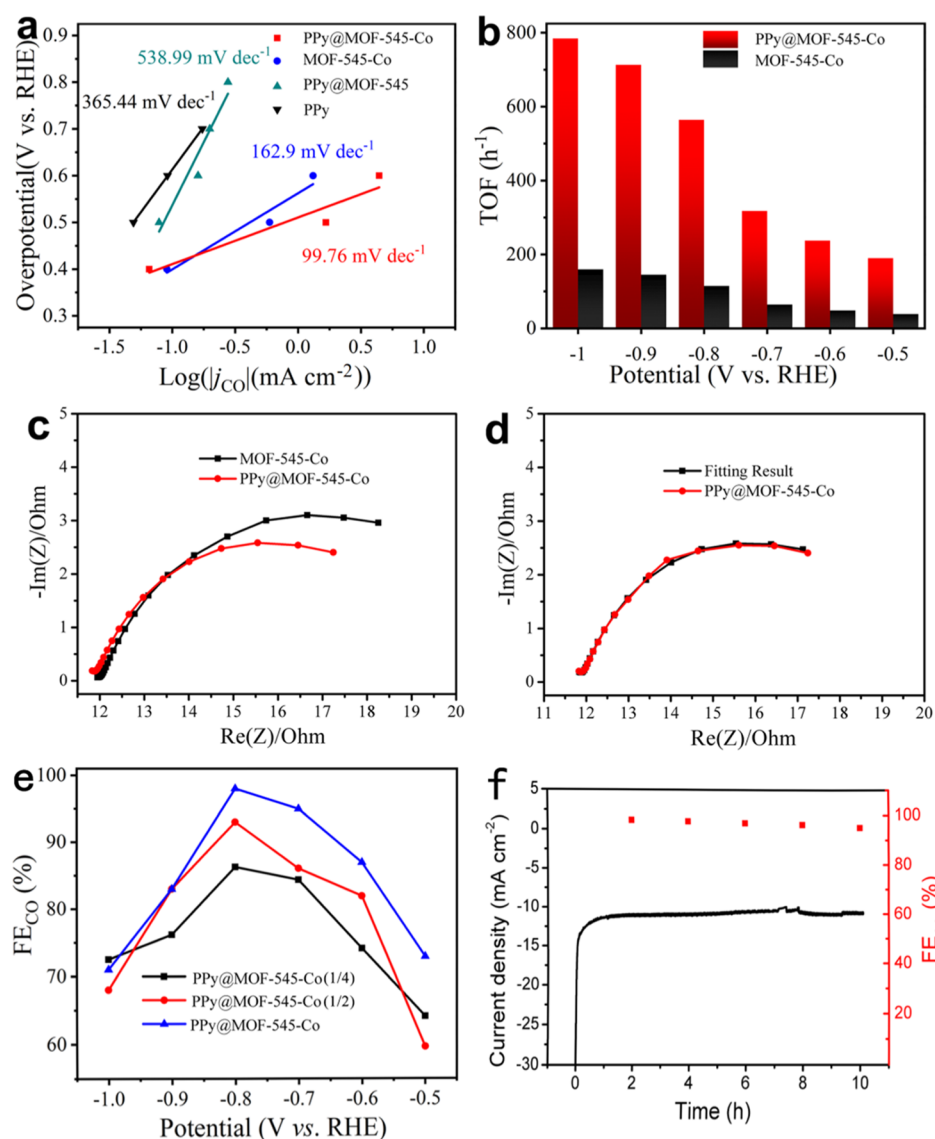


**Figure 3.** Electrocatalytic CO<sub>2</sub>RR performance of PPy@MOF-545-Co. (a) LSV curves of PPy@MOF-545-Co and contrastive samples. (b) LSV curves of PPy@MOF-545-Co at CO<sub>2</sub>- and Ar-saturated 0.1 M KHCO<sub>3</sub>. (c) FE<sub>CO</sub> of PPy@MOF-545-Co and contrastive samples measured under different voltages. (d) Partial CO current density. (e) Summary of electrocatalytic performances of the literature-reported materials and metal-doped porphyrin MOFs.

we choose PPy-loaded MOF-545-Co as a desired platform for further investigation.

To further investigate the specific catalytic performance of PPy@MOF-545-Co, a series of comparison samples [i.e., MOF-545, MOF-545-Co, PPy, and physical mixture (MOF-545-Co/PPy with a ratio of 83/17)] were prepared to test their performances (Figure S12). For these samples, MOF-545-Co (FE<sub>CO</sub> = 57%, −0.8 V) and MOF-545-Co/PPy (62%, −0.8 V) exhibited lower catalytic performance than PPy@MOF-545-Co (98%, −0.8 V), which could be ascribed to the lower electron-transfer ability. On the contrary, MOF-545 and PPy exhibited a negligible catalytic activity for CO<sub>2</sub> reduction, which might be explained by the absence of Co active sites that resulted in a sharp decrease of catalytic activity. It also implies that the notable electrocatalytic performance of PPy@MOF-545-Co originates from the cooperation effect between Co-TCPP and PPy molecules. To further study the effect of PPy on the catalytic performance of MOF, we tested the CO<sub>2</sub>RR catalytic performance of PPy@MOF-545-Co with different amounts of PPy. To investigate the effect of PPy, we applied three diverse

loading amounts of PPy in MOF-54-Co to test the catalytic performance. We obtained the hybrid MOF materials with pure Py (denoted as PPy@MOF-545-Co), half the amount of Py [denoted as PPy@MOF-545-Co (1/2)], and quarter the amount of Py [denoted as PPy@MOF-545-Co (1/4)], respectively. Figure 4e shows the FE<sub>CO</sub> for MOF with different Py loadings, and the result indicated that the catalytic performance increased with the enhancement of Py loading. This might be because pure Py loaded in MOF channels can polymerize into continuous PPy molecules. When the concentration of Py decreased, the shorter PPy chain might be formed. Under these circumstances, it is difficult to form a continuous electron transport medium, thus resulting in lower catalytic performance. As seen in Figure 4e, when the PPy loading amount decreases, FE<sub>CO</sub> gradually decreases from 98 (PPy@MOF-545-Co) to 87% [PPy@MOF-545-Co (1/4)]. In addition, the calculated turnover frequency (TOF) of PPy@MOF-545-Co was up to about 785 h<sup>-1</sup> at a potential of −1.0 V (vs hydrogen evolution reaction), which is superior to that of MOF-545-Co and other contrast samples (Figure 4b).



**Figure 4.** Electrocatalytic CO<sub>2</sub>RR performances of PPy@MOF-545-Co and contrast samples. (a) Tafel plots of PPy@MOF-545-Co and contrastive samples. (b) TOF of PPy@MOF-545-Co and MOF-545-Co at different potentials. (c) Nyquist plots of PPy@MOF-545-Co and MOF-545-Co. (d) PPy@MOF-545-Co fitting result over the frequency ranging from 1000 kHz to 0.1 Hz at −0.8 V vs RHE. (e) FE<sub>CO</sub> of PPy@MOF-545-Co, PPy@MOF-545-Co (1/2), and PPy@MOF-545-Co (1/4) at different potentials. (f) Durability test of PPy@MOF-545-Co at a potential of −0.8 V vs RHE.

To evaluate the electrocatalysis stability, the durability of CO<sub>2</sub>RR has been measured. The current density of PPy@MOF-545-Co remained almost constant (about −11 mA cm<sup>−1</sup>), and FE<sub>CO</sub> maintained 95% of the initial FE for CO production after 10 h of continuous electrolysis at −0.8 V (Figure 4f). Besides, the PXRD pattern indicated that the structure remained unchanged after 10 h of the electrocatalytic reaction, which implied high stability during the electrocatalytic reaction (Figure S13).

Tafel slope was an important parameter often used to characterize the performance of an electrochemical catalyst as it quantified the amount of additional applied potential required to observe the logarithmic increase in the measured current.<sup>50</sup> The smaller the slope of the Tafel curve, the lower the overpotential of the catalytic process under the same kinetic current density or apparent current density. The Tafel slopes of PPy@MOF-545-Co was 99.8 (mV dec<sup>−1</sup>), which was obviously smaller than those of MOF-545-Co (162.9 mV

dec<sup>−1</sup>), PPy@MOF-545 (539.0 mV dec<sup>−1</sup>), and PPy (365.4 mV dec<sup>−1</sup>), suggesting that PPy@MOF-545-Co has the most favorable kinetics for the reduction of CO<sub>2</sub> and excellent catalytic activity (Figure 4a). Electrochemical impedance spectroscopy was generally applied to analyze the structure of the electrochemical system and the properties of the electrode process. As shown in the Nyquist plots (Figure 4c), the charge-transfer resistance (*R*<sub>ct</sub>) of PPy@MOF-545-Co was 7.5 Ω, which was lower than that of MOF-545-Co (12.5 Ω), demonstrating that PPy@MOF-545-Co possessed much lower charge-transfer resistance in the electron-transfer process. The electron conductivity test further proved that the loading of PPy enhanced the electron conductivity of MOF-545-Co to a certain extent (Figure S9). This would be attributed to the possible interactions (e.g.,  $\pi$ – $\pi$  interaction or van der Waals force) between PPy molecules and Co-TCPP center of MOF-545-Co might enhance the charge-transfer ability for the active site in the CO<sub>2</sub>RR process.

## CONCLUSIONS

In summary, a feasible low-temperature in situ polymerization method has been applied to impart oriented PPy chains in the pores of MOF-545-Co. This method not only preserves the catalytic sites in the host framework but also improves the electrochemical conductivity of the materials, which is much beneficial for the improvement of CO<sub>2</sub>RR performance. Specifically, the FE<sub>CO</sub> of PPy@MOF-545-Co reaches as high as 98% at −0.8 V, almost 2 times higher than that of bare MOF-545-Co. The high performance might be attributed to the incorporation of PPy that can serve as electric cables in the channel of MOF that facilitate electron transfer during the CO<sub>2</sub>RR process. This method will pave a promising pathway for the design of high-efficiency catalysts and should inject new vitality not only to CO<sub>2</sub> reduction research but also to the extensive electrocatalytic field.

## ASSOCIATED CONTENT

### Supporting Information

The Supporting Information is available free of charge at <https://pubs.acs.org/doi/10.1021/acsami.1c15187>.

Details of synthesis; material characterizations; XRD patterns; additional SEM, TEM images; XPS spectra; and electrochemical characterizations (PDF)

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Z.X., Y.C., and Y.-Q.L. instructed this work and cowrote this paper. J.L., X.W., and Z.Y. carried out the experiments. K.S. contributed to the GC spectrum measurement.

## Notes

The authors declare no competing financial interest.

## ACKNOWLEDGMENTS

This work was financially supported by the National Natural Science Foundation of China (nos. 22171139, 21471003, 21871141, 21871142, 21701085, and 21901122); Natural Science Foundation of Educational Commission of Anhui Province of China (no. KJ2020A0240); the Natural Science Research of Jiangsu Higher Education Institutions of China (no. 19KJB150011); and Project funded by China Postdoctoral Science Foundation (no. 2019M651873).

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