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Redox Molecular Junction Metal-Covalent Organic Frameworks for Light-assisted CO₂ Energy Storage

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Abstract: Visible-light sensitive and bi-functionally favored CO₂ reduction (CRR)/evolution (CER) photocathode catalysts that can get rid of the utilization of ultraviolet light and improve sluggish kinetics is demanded to conquer the current technique-barrier of traditional Li-CO₂ battery. Here, a kind of redox molecular junction sp²c metal-covalent organic framework (i.e. Cu₃-BTDE-COF) has been prepared through the connection between Cu₃ and BTDE and can serve as efficient photocathode catalyst in light-assisted Li-CO₂ battery. Cu₃-BTDE-COF with redox-ability, visible-light-adsorption region, electron-hole separation ability and endows the photocathode with excellent round-trip efficiency (95.2%) and an ultralow voltage hysteresis (0.18 V), outperforming the Schiff base COFs (i.e. Cu₃-BTDA-COF and Cu₃-DT-COF) and majority of the reported photocathode catalysts. Combined theoretical calculations with characterizations, Cu₃-BTDE-COF with the integration of Cu₃ centers, thiazole and cyano groups possess strong CO₂ adsorption/activation and Li⁺ interaction/diffusion ability to boost the CRR/CER kinetics and related battery property.

Introduction

Nowadays, the energy consumption is still dominated by non-renewable fuels such as fossil fuels and the challenge of insufficient energy supply is becoming increasingly prominent, thus it is crucial to explore innovative and substitutable energy sources [1-4]. Aprotic Li-CO₂ battery based on the reversible electrochemical CO₂ reduction (CRR) and evolution (CER) reactions offers an CO₂ fixation approach as well as energy storage for diverse applications like power plant or Mars exploration [5-8]. Specially, it possesses a superior theoretical energy density (1876 Wh kg⁻¹) to Li-ion battery based on the reversible reaction: 4Li + 3CO₂ ⇌ 2Li₂CO₃ + C (E₀ = 2.80 V vs. Li⁺/Li [5, 9-11]). To date, plenty of nanomaterials (e.g., noble metals, carbon materials and transition metal oxides) have been applied in Li-CO₂ batter [12-15]. However, there are some bottlenecks restrict its practical applications, such as high overpotential, poor round-trip efficiency or low-rate performance, etc., which mainly come from the slow charge/discharge reaction kinetic [16, 17]. In most of Li-CO₂ battery systems, the sluggish reaction kinetics especially that caused by the insulation or low decomposable properties of Li₂CO₃ during the discharge/charge processes lead to large voltage gap (mostly >1 V) and poor reversible ability [18,

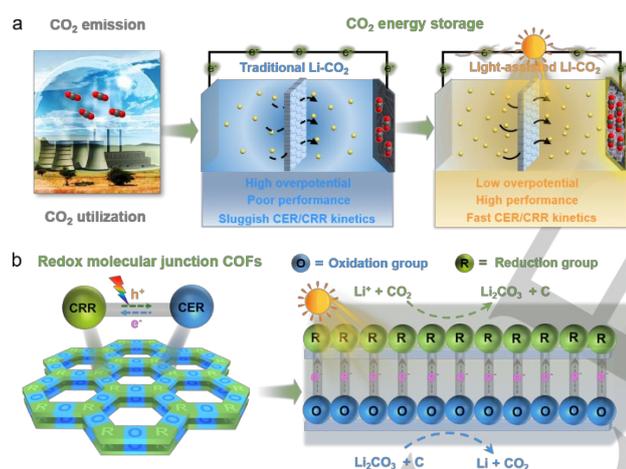
19]. The difficulty in decomposing discharge products calls for advanced auxiliary technologies to overcome technical barriers of traditional Li-CO₂ battery, yet it remains a daunting challenge to achieve this goal [20, 21].

Light energy, an external driving force to promote the electrode reactions, has been highlighted as an attractive and alternative strategy for Li-CO₂ batter [22]. During the operation of light-assist Li-CO₂ battery, a certain amount of incident light can excite a large number of photo-generated electrons and holes on catalysts based on the photoelectric effect to promote the CRR and CER, thus reducing voltage hysteresis and improving battery efficiency [6, 22-24]. During the past years, only a few cathode catalysts such as In₂S₃@CNT/SS [23], CNT@C₃N₄ [24], SiC@RGO [25], CoPc-Mn-O [6], or TiO₂/CC [26], have been investigated in light-assisted Li-CO₂ battery. However, they still face drawbacks like: i) the light absorption for most of the generally applied semiconductors are still in the ultraviolet region [26]; ii) the study of mechanisms for the light-assisted interaction are still rare and iii) most of materials are based on nonporous or hybrid materials, which are difficult to make full use of the active sites or precisely modulate their inherent structures [6, 23, 25]. Therefore, it is meaningful to specially develop photocatalysts with broad light absorption range, high porosity or well-designed functional groups to accelerate the reaction kinetics of CRR/CER to meet the demands of advanced light-assisted Li-CO₂ battery.

In this regard, covalent organic frameworks (COFs), a kind of crystalline porous polymers connected by strong covalent bond, come to our mind as a promising candidate for light-assisted Li-CO₂ battery [27-30]. Since 2005, COFs with the properties of pre-desirable structures/functions, high stability and porosity have attracted a lot of attention in various application fields like gas adsorption/separation, sensing, catalysis, drug delivery or energy storages, et [31, 32]. Among these advantages, the most remarkable one is their highly tunable and modifiable structures that can be accurately constructed through the selection of well-designed struts to meet the requirements of advanced applications like Li-CO₂ battery. Metal-covalent organic frameworks (MCOFs), a kind of interesting COFs obtained through the covalent bonding between metal clusters and organic ligands, possesses combined properties and advantages of both metal-organic frameworks (MOFs) and COFs, in which the metal clusters simulated from MOFs can serve as active sites while the covalent connection would be highly tolerant for many potentially

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harsh application^[33, 34]. Especially for the covalent connection obtained through low-reversible vinylene linkages, the resultant sp^2 carbon-conjugated COFs (sp^2c -COFs) might present high chemical stability, in-plane conjugations, functional -CN groups and porous frameworks that are much attractive for energy related applications^[35-37]. Thus, if the functions of sp^2c -COF and MCOF can be well-combined, the generated sp^2c -MCOF might serve as very promising candidates to serve as ideal light-assisted Li-CO₂ photocathode catalysts due to the following advantages: i) COFs can be photo-absorbent by introducing photosensitive groups; ii) the adjustable structures would endow MCOFs with efficient ligand-metal charge transfer (LMCT) effect for enhanced electron-hole separation efficiency that might serve as redox molecular junction units for CRR and CER^[38, 39]; iii) the carbon-carbon double bonds and additionally existed -CN groups in sp^2c -MCOF enhance the π conjugated electron transition, which might increase the electron mobility or Li⁺ interaction^[40-42] iv) MCOFs with ordered and well-defined structures are beneficial for mechanism study. Up to date, their applications in light-assisted Li-CO₂ battery remain an unexplored field and need much efforts to be made. Therefore, it is desirable to design and synthesize sp^2c -MCOF with multifunctional active sites to meet the requirements of CRR and CER for high performance light-assisted Li-CO₂ battery.



Scheme 1. Schematic representation of redox molecular junction COFs for light-assisted CO₂ energy storage. (a) The comparison of traditional Li-CO₂ battery and light-assisted Li-CO₂ battery. (b) The advantages of redox molecular junction COFs in light-assisted Li-CO₂ battery.

Herein, a kind of redox molecular junction sp^2c -MCOF (i.e. Cu₃-BTDE-COF) with multiple active sites are synthesized by in-situ covalent connection of Cu cluster (Cu₃), 2,2'-(benzo[c][1,2,5]thiadiazole-4,7-diylbis(4,1-phenylene)diacetonitrile (BTDE) that can serve as desired platform to be applied in light-assisted Li-CO₂ battery (Scheme 1). Specifically, the covalent connection between Cu₃ and BTDE in Cu₃-BTDE-COF enables the visible light driven electron-hole to efficiently separate and transfer from BTDE (oxidation group) to Cu₃ (reduction group) moiety, resulting in the photoexcited electrons on Cu₃ and holes on BTDE, respectively. Notably, Cu₃-BTDE-COF based cell offers an ultralow voltage hysteresis (0.18 V), outperforming Schiff base COFs (i. e. Cu₃-BTDA-COF and

Cu₃-DT-COF), which is one of the best photocathode reported to date (Table S1 and S2). In addition, theoretical calculations and characterization have shown that Cu₃-BTDE-COF with the integration of metal active centers, thiazole and cyano multifunctional groups possesses strong CO₂ adsorption/activation and Li⁺ interaction/diffusion ability, thereby improving overall battery efficiency.

Results and Discussion

Structural design and fabrication of photocatalysts. Cu₃-BTDE-COF has been synthesized by the condensation reaction of [Cu₃(PyCA)₃] (Cu₃) and 2,2'-(benzo[c][1,2,5]thiadiazole-4,7-diylbis(4,1-phenylene)diacetonitrile (BTDE) (detail see Methods). A kind of reddish-brown powder can be obtained after 72 h reaction at 150 °C and the structure of the sample is confirmed by the powder X-ray diffraction (PXRD) combined with structural simulation (Figure 1a and Figure S1). On the basis of theoretical structural simulation and Pawley refinements, the Cu₃-BTDE-COF can be fitted into the ABC stacking model with the corresponding unit cell parameters of $a = 48.00 \text{ \AA}$, $b = 48.00 \text{ \AA}$, and $c = 2.63 \text{ \AA}$, and $\alpha = \beta = 90^\circ$, $\gamma = 120^\circ$ in the R_3 space group (Table S3). As revealed by the differential plot, the refinement of PXRD pattern is well agreement with the experimental one (unweighted-profile R factor (R_p), 2.62% and weighted profile R factor (R_{wp}), 3.38%), suggesting the validity of the computational model (Figure 1c and e). In the PXRD pattern, it has intense peaks at 3.68° , 6.39° , 7.36° , and 10.13° , which can be ascribed to the (1 1 0), (3 0 0), (2 2 0) and (2 1 1) facets of Cu₃-BTDE-COF, respectively. In addition, Fourier transform infrared spectroscopy (FTIR) has been applied to characterize the chemical compositions of Cu₃-BTDE-COF. In the FTIR spectra, the stretching vibration at 1620 cm^{-1} can be attributed to the C=C in the resultant Cu₃-BTDE-COF, suggesting the successful formation of Cu₃-BTDE-COF. Meanwhile, this result can be further certified by the obviously decreased -CHO stretching vibration band (1699 cm^{-1}). Besides, the cyano group stretching vibration at 2220 cm^{-1} belongs to BTDE of Cu₃-BTDE-COF^[43, 44].

In addition, X-ray photoelectron spectroscopy (XPS) analysis has been carried out to investigate the chemical states of the existed elements in Cu₃-BTDE-COF. For the XPS spectra of Cu₃-BTDE-COF, five main peaks with binding energy of 933.85, 164.65, 284.64, 399.10 and 531.55 eV are attributed to Cu 2p, S 2p, C 1s and O 1s, respectively (Figure S2). The C 1s spectra display two peaks with binding energy at 286.35 and 284.70 eV, respectively. The peak at 286.35 eV is assigned to the cyano group, and the peak at 284.70 eV belongs to the C-C/C=C bonds of the Cu₃-BTDE-COF^[45]. In the Cu 2p region, peaks of 935.13 and 933.65 eV are assigned to Cu²⁺ and Cu⁺, respectively, which complies previously reported works^[46]. In detail, the appearance of Cu²⁺ is due to the surface oxidation, and the Cu valence can be switched between Cu⁺ and Cu²⁺ without altering the components of Cu₃. In addition, the XPS spectra of S, C and O elements of Cu₃-BTDE-COF are shown in Figure S4. In the S 2p region, peaks at 164.07 eV (S 2p_{3/2}) and 165.29 eV (S 2p_{1/2}) can be attributed to the characteristic peaks of N-S bond in BT monomer (Figure S2). Besides, its N 1s spectra displays a single peak at 398.40 eV, corresponding to the C=N/C-N of Cu₃-BTDE-COF^[47].

Additionally, Cu₃-BTDA-COF and Cu₃-DT-COF with 4,4'-(benzo-2,1,3-thiadiazole-4,7-diyl)dianiline (BTDA) and 4,4'-

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diamino-triphenyl (DT) substituted BTDE of Cu₃-BTDE-COF have also been synthesized as contrastive samples of Cu₃-BTDE-COF (Table S4 and S5). For Cu₃-BTDA-COF and Cu₃-DT-COF, their experimental PXRD patterns show high crystallinity, which are in good agreement with their simulated ones (Figure 1b and d). In addition, the compositions of them have also been characterized by FTIR and XPS tests (Figure S3-S8). It can be seen from the above test results that Cu₃-BTDA-COF and Cu₃-DT-COF have been successfully synthesized that can serve as ideal contrast samples to be compared with Cu₃-BTDE-COF.

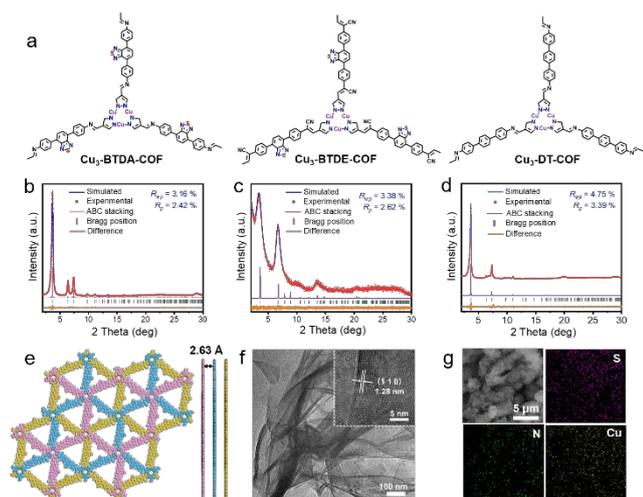


Figure 1. Structure and characterization of Cu₃-BTDE-COF, Cu₃-BTDA-COF and Cu₃-DT-COF. (a) Schematic representation of the syntheses of Cu₃-BT-COF, Cu₃-pT-COF and TFP-BT-COF. (b-d) Experimental (red dot) and simulated (blue line) PXRD patterns of Cu₃-BTDE-COF (b), Cu₃-BTDA-COF (c) and Cu₃-DT-COF (d). (e) Top and side views of the structure of Cu₃-BTDE-COF. (f) The HRTEM image and lattice fringes of Cu₃-BTDE-COF (inset shows the circled place in f). (g) SEM and elemental mapping images of Cu₃-BTDE-COF.

The porosity of Cu₃-BTDE-COF has been evaluated by N₂ sorption tests at 77 K (Figure S9). The Brunauer-Emmett-Teller surface area (S_{BET}) of Cu₃-BTDE-COF is calculated to be 151 m² g⁻¹. To evaluate the CO₂ adsorption ability of Cu₃-BTDE-COF, the CO₂ adsorption test is carried out and it presents CO₂ uptake capacity of 33.7 cm³ g⁻¹ and 12.5 cm³ g⁻¹ at 273 K and 298 K, respectively (Figure S10). Moreover, the morphology of the synthesized Cu₃-BTDE-COF have been studied by transmission electron microscopy (TEM) and scanning electron microscope (SEM). Figure 1f shows the typical TEM image of Cu₃-BTDE-COF, demonstrating a layer-stacked morphology. Specifically, the oriented lattice fringes of Cu₃-BTDE-COF are clearly visible and the detected lattice spacing of 1.28 nm can be ascribed to the (1 1 0) plane in the high-resolution TEM (HR-TEM) image, proving the high crystallinity of Cu₃-BTDE-COF (Figure 1f). Besides, the elemental mapping analysis displays that Cu, N and S are homogeneously distributed in Cu₃-BTDE-COF (Figure 1g). The above morphology characterizations further confirm that Cu₃-BTDE-COF has good crystallinity and long-range ordering structure. In addition, the morphology of Cu₃-BTDA-COF and Cu₃-DT-COF has also been characterized by SEM and TEM tests (Figure S11-S14).

Characterization of photocatalysts. In order to improve the electrochemical property, the photocathode for light-assisted Li-CO₂ battery needs to exhibit effective utilization of the light and low recombination rate for the photogenerated electron-hole pairs

[48]. Therefore, the light absorption ability, electrons-holes separation efficiency and energy band structure of the sample have been analyzed. Initially, the optical absorption performances of Cu₃-BTDE-COF, Cu₃-BTDA-COF and Cu₃-DT-COF are carried out by ultraviolet and visible (UV-Vis) absorption spectra (Figure 2a). The band gaps (E_g) of Cu₃-BTDE-COF, Cu₃-BTDA-COF and Cu₃-DT-COF are determined to be 2.13, 2.01 and 2.69 eV by their Tauc plots, indicating that they all possess the properties of semiconductors (Figure S15) [6, 23]. In addition to these, Mott-Schottky measurements have also been performed to study the band positions of Cu₃-BTDE-COF, Cu₃-BTDA-COF and Cu₃-DT-COF (Figure 2c). The lowest unoccupied molecular orbital (LUMO) of the Cu₃-BTDE-COF is calculated to be 2.09 V versus Li⁺/Li. Meanwhile, the highest occupied molecular orbital (HOMO) is calculated to be 3.89 V. As shown in Figure 2d, Cu₃-BTDE-COF fulfills the essential requirement of the light-assisted Li-CO₂ battery that the redox potential of CO₂/Li₂CO₃ (2.80 V vs Li⁺/Li) is located between the LUMO and HOMO [49, 50].

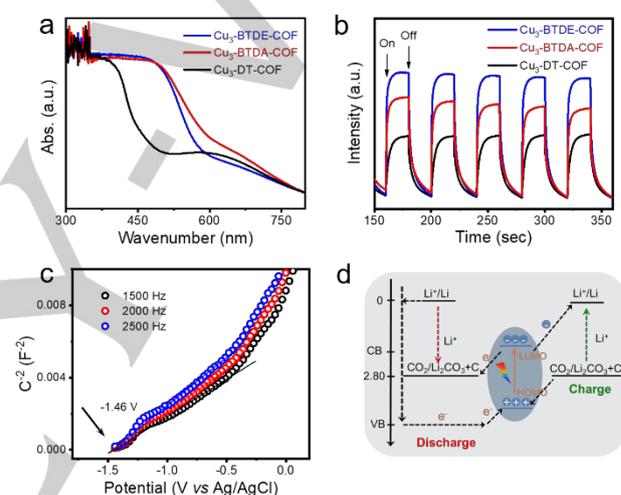


Figure 2. Characterizations of the optical properties for Cu₃-BTDE-COF, Cu₃-BTDA-COF and Cu₃-DT-COF. (a) UV-Vis absorption spectra. (b) Transient photocurrent response. (c) Mott-Schottky plots of Cu₃-BTDE-COF. (d) Energy diagram of Cu₃-BTDE-COF and standard potential of CO₂/Li₂CO₃ versus Li⁺/Li.

Thus, Cu₃-BTDE-COF enables the sufficient utilization of photogenerated electrons/holes for CRR and CER. Moreover, we also have carried out the UV-Vis spectra and Mott-Schottky tests on Cu₃-BTDA-COF and Cu₃-DT-COF (Figure S16 and S17). According to the tests results, the conduction band potentials (1.49 V vs Li⁺/Li) and valence band potentials (3.21 V and 4.01 V vs Li⁺/Li) of Cu₃-BTDA-COF and Cu₃-DT-COF can also match with the redox potential (2.80 V vs Li⁺/Li) of CO₂/Li₂CO₃. These results verify that both Cu₃-BTDA-COF and Cu₃-DT-COF can use light energy to complete the CRR and CER processes. In addition, the transient photocurrent response has been performed to study the charge transfer behaviors (Figure 2b). The intensity of transient photocurrent response for Cu₃-BTDE-COF is much higher than those of Cu₃-BTDA-COF and Cu₃-DT-COF (Figure 2b). Generally speaking, the enhancement of the photo-responsive of catalysts has a positive impact on their photocatalytic performance. Compared with Cu₃-BTDA-COF and Cu₃-DT-COF, Cu₃-BTDE-COF can theoretically generate more photoinduced electrons for CRR because of the specific π -electron delocalization due to its sp² carbon linkage^[50].

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In addition, we have carried out the electron paramagnetic resonance (EPR) tests to study its photogenerated electrons/holes transfer behavior. Under light conditions, the EPR signals of Cu₃-BTDE-COF, Cu₃-BTDA-COF and Cu₃-DT-COF are significantly higher than that under dark conditions at $g=2.003$ (Figure S18-S20). The result shows that the Cu₃-BTDE-COF, Cu₃-BTDA-COF and Cu₃-DT-COF can efficiently accelerate the electron-hole separation and prevent recombination^[51]. Even though Cu₃-BTDE-COF, Cu₃-BTDA-COF, and Cu₃-DT-COF all have signals in the EPR tests. Furthermore, the intensity of EPR signals of Cu₃-BTDE-COF and Cu₃-BTDA-COF are stronger than that of Cu₃-DT-COF under the dark and light conditions at $g=2.003$ (Figure S19 and S20), which shows that thiazole is an important photosensitive group. The above results illustrate the photosensitivity of Cu₃-BTDE-COF, Cu₃-BTDA-COF and Cu₃-DT-COF, which may greatly benefit their application in light-assisted Li-CO₂ battery. Combining the ultraviolet photoelectron spectroscopy and Tauc curves, the LUMO and HOMO of Cu₃-BTDE-COF are calculated to be 1.84 V and 4.41 V vs. Li⁺/Li, respectively (Figure S15). On account of the LUMO potential is lower than the reduction potential between CO₂ and Li₂CO₃/C (2.80 V vs. Li⁺/Li), the photoelectrons can be applied in the discharge reactions of Li-CO₂ battery. Similarly, the holes can promote the charge reactions of Li-CO₂ battery in reverse charging process, because the HOMO is higher than the oxidation potential between Li₂CO₃/C and CO₂ (2.80 V vs. Li⁺/Li). These results indicate that the efficient generation, separation, transfer, and utilization of photogenerated electrons and holes in the target cell process are promoted by the Cu₃-BTDE-COF photocathode.

Light-assisted Li-CO₂ battery. The electrochemical performances of Cu₃-BTDE-COF, Cu₃-BTDA-COF and Cu₃-DT-COF have been systematically evaluated as photocathodes of Li-CO₂ battery. Figure 3a, Figure S22 and S24 show the galvanostatic discharge/charge profiles of Li-CO₂ battery based on Cu₃-BTDE-COF, Cu₃-BTDA-COF and Cu₃-DT-COF photocathodes with/without illumination at 200 mA/g. The Cu₃-BTDE-COF photocathode under the dark condition displays discharge and charge voltages of 2.7 V and 4.1 V, respectively, with a lower round-trip efficiency of 65.8%. Under the light conditions, the discharge voltage of Cu₃-BTDE-COF photocathode increases to 3.32 V and the charging voltage decreases to 3.5 V, showing a high round-trip efficiency of 95.2%, indicating excellent light responsiveness of Cu₃-BTDE-COF photocathode (Figure 3a). In comparison, the discharge voltage and charge voltage of Cu₃-BTDA-COF photocathode under the dark conditions are 2.64 V and 4.2 V, respectively, corresponding to lower round-trip efficiency (62.85%). Under light conditions, the charge and discharge voltage are 3.65 V and 3.29 V, respectively, and the round-trip efficiency also increases to 90.13%. Cu₃-DT-COF presents the worst performance among them as a light-assisted Li-CO₂ battery under both dark (charge voltage, 4.31 V and discharge voltage, 2.56 V) and the light condition (charge voltage, 3.61 V and discharge voltage, 3.23 V). It has been reported that Cu₃ is a CO₂ reducing group, and Cu₃-DT-COF obtained through the covalent connection of Cu₃ and DT has faster charge transfer ability and higher absorbance compared with Cu₃^[4]. Meanwhile, the energy round-trip efficiency of Cu₃-DT-COF under the light conditions is improved by about 30% compared to that under the dark conditions (Figure S21 and S22). Thus, Cu₃-BTDE-COF has the strongest response to light, compared to Cu₃-BTDA-COF and Cu₃-DT-COF, which may be

due to the versatility produced by the many functional groups in Cu₃-BTDE-COF. After that, we have tested the rate capability at different current densities and the results showed that it could maintain low overpotentials at different current densities (Figure S23).

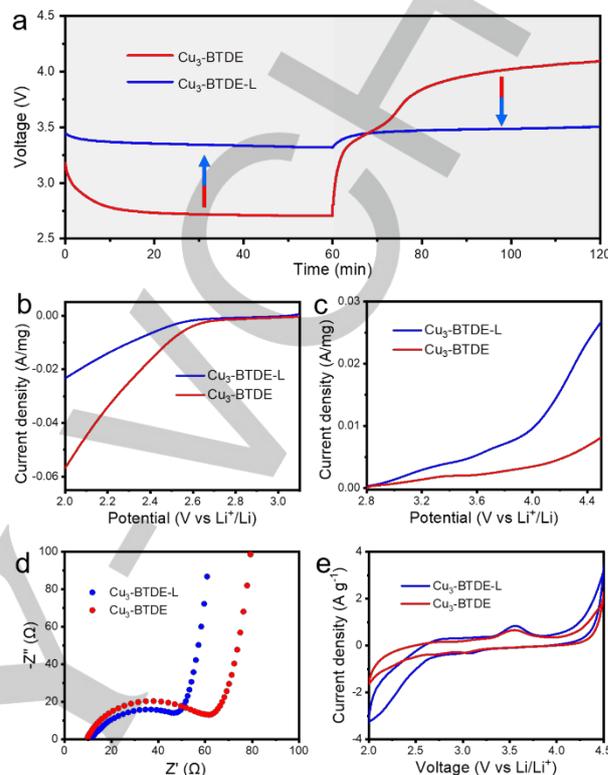


Figure 3. Battery performance of Cu₃-BTDE-COF. (a) Discharge and charge curves of the Cu₃-BTDE-COF based Li-CO₂ battery with and without illumination at 200 mA/g. (b) LSV curves of Cu₃-BTDE-COF cathode during CRR process. (c) LSV curves of Cu₃-BTDE-COF cathode during CER process. (d) EIS images of Cu₃-BTDE-COF with and without illumination at 200 mA/g. (e) CV curves of the Cu₃-BTDE-COF based cathode with and without illumination.

To gain insight into the influence of light on the Li-CO₂ battery, we have measured linear sweep voltammetry (LSV) curves in a CO₂-saturated 1.0 M LITFSI in tetraethylene glycol dimethyl ether (TEGDME) environment (Figure 3b and 3c). Under light conditions, the onset potentials and the limiting current density for the CRR and CER of Cu₃-BTDE-COF under light are much higher than those of Cu₃-BTDE-COF without light. The cyclic voltammetry (CV) results further confirm that the activity of the system has been improved with the assistance of light energy (Figure 3e). The performance test results of Li-CO₂ battery fully demonstrate their excellent properties. Thus, we have carried out cycling tests by Cu₃-BTDE-COF photocathode with and without illumination at 200 mA/g (Figure S24). These results indicate that the discharge and charging process of Cu₃-BTDE-COF in photo-assisted Li-CO₂ battery can be promoted by photogenerated electrons and holes, thus ensuring the cycling stability of Cu₃-BTDE-COF photoelectrode during the long-term cycling test. In addition, the theoretical thermodynamic potential of Li₂CO₃/C to CO₂/Li⁺ is 2.80 V vs. Li⁺/Li, meanwhile the value is 3.82 V vs. Li⁺/Li for the self-decomposition of Li₂CO₃.²³ When the battery is cycled under light condition, the charging voltage is basically lower than 3.82 V, indicating that the charging process

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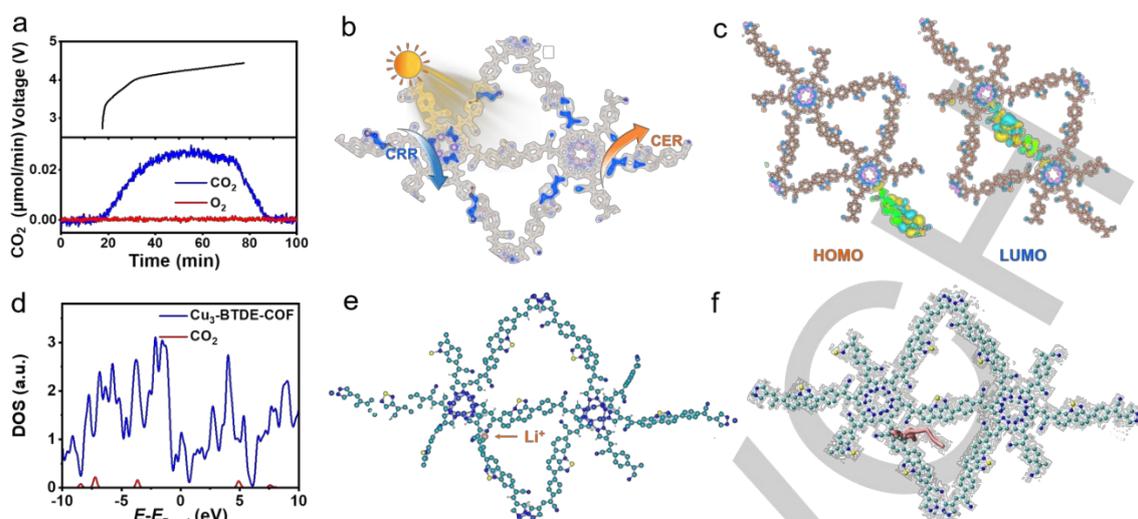


Figure 4. DFT calculations of $\text{Cu}_3\text{-BTDE-COF}$ in light-assisted Li-CO_2 battery. (a) DEMS test during charging of the Li-CO_2 battery with $\text{Cu}_3\text{-BTDE-COF}$ cathode catalyst. (b) Differential charge density diagram of $\text{Cu}_3\text{-BTDE-COF}$. (c) The associated HOMO and LUMO of $\text{Cu}_3\text{-BTDE-COF}$. (d) The density of states for $\text{Cu}_3\text{-BTDE-COF}$ in light-assisted Li-CO_2 battery. (e) Theoretical calculation model of Li^+ with $\text{Cu}_3\text{-BTDE-COF}$. (f) The diffusion pathway for Li^+ to migrate on the structure model of $\text{Cu}_3\text{-BTDE-COF}$.

is more likely to be the co-decomposition of Li_2CO_3 and carbon species, rather than the self-decomposition of Li_2CO_3 . During the operation of the battery, the inevitable accumulation of photocathode discharge products and the volatilization of liquid components in the electrolyte result in the decrease of cycle efficiency. In contrast, the batteries under dark condition are constantly subjected to severe polarization voltages, which can lead to adverse decomposition of battery materials or by-product formation [6, 24]. Afterwards, we have carried out the electrochemical impedance spectra (EIS) test to analyze the superior performance of $\text{Cu}_3\text{-BTDE-COF}$ photocathode (Figure 3d). The semicircle of $\text{Cu}_3\text{-BTDE-COF}$ under the light conditions is noticeably smaller than that of $\text{Cu}_3\text{-BTDE-COF}$ under the dark conditions, proving the more excellent charge transfer kinetics of $\text{Cu}_3\text{-BTDE-COF}$ in light-assisted Li-CO_2 battery [52]. These results have indicated that the introduction of light energy can indeed improve the charging and discharging performance of Li-CO_2 battery. We also conducted LSV and CV tests on $\text{Cu}_3\text{-BTDA-COF}$ and $\text{Cu}_3\text{-DT-COF}$ under light and dark conditions, respectively (Figure S25-S28), proving that $\text{Cu}_3\text{-BTDA-COF}$ and $\text{Cu}_3\text{-DT-COF}$ have stronger redox properties under light conditions.

Moreover, a series of tests have been conducted to study the by-products during the discharge/charge processes. Specifically, we have carried out the PXRD tests to determine the discharge products, and the main discharge products is Li_2CO_3 with or without light (Figure S29 and S30). However, it is clear from Figure S30 that Li_2CO_3 has lower crystallinity under light conditions, which means that the decomposition process becomes easier compared to dark conditions. In order to elucidate the work mechanism of light field and strong electric field on the battery reactions, the discharged/recharged $\text{Cu}_3\text{-BTDE-COF}$ cathodes under the dark and light conditions were characterized. Then, we have performed SEM tests to study the different battery stages. From the SEM images, it can be seen that the morphology of the product is different under light and dark conditions. Under dark conditions, the detected discharge products are rod-shaped nanoparticles with sizes of $\sim 1 \mu\text{m}$ (Figure S31). Under light

conditions, the morphology of the discharge products exhibits nanoflake morphology with a size of only $\sim 100 \text{ nm}$, in significant contrast to that under dark condition (Figure S31). In order to further prove the complete reversibility of $\text{Cu}_3\text{-BTDE-COF}$ photocathode of the battery, we have carried out in-situ differential electrochemical mass spectrometry (DEMS) test to evaluate the gas evolution during the charge process (Figure 4a). As shown in Figure 4a, DEMS testing can monitor the gas release of $\text{Cu}_3\text{-BTDE-COF}$ cathode batteries during charging (Figure 4a), and only CO_2 gas has been detected. Based on the DEMS results, the charge to mass ratio ($4.21 \text{ e}^-/3\text{CO}_2$) during the charging process can be determined, meaning that it basically matches with the reversible reaction of $4\text{Li}^+ + 3\text{CO}_2 + 4\text{e}^- \rightarrow 2\text{Li}_2\text{CO}_3 + \text{C}$. These results indicate that the battery performance can be effectively increased due to the synergistic effect of $\text{Cu}_3\text{-BTDE-COF}$ photocathode catalyst.

Based on the above experiments results, theoretical calculations have been carried out to investigate the possible working mechanism of $\text{Cu}_3\text{-BTDE-COF}$ photocathode in light-assisted Li-CO_2 battery. Initially, we have performed differential charge density test to analyze the charge distribution of $\text{Cu}_3\text{-BTDE-COF}$ structure. As shown in Figure 4b, the interfacial charge density difference is depicted, in which the charge accumulation is presented as the blue region of Cu atoms on Cu_3 , and the charge depletion is the gray region close to thiazole on BTDE (Figure 4b). This result shows that the electrostatic potential value around Cu_3 is positive, suggesting that it is more electrophilic than that of BTDE. Therefore, the result complies with above-mentioned experimental results that CRR and CER are more likely to locate in Cu_3 and BTDE, respectively (Figure 4b). To support this result, we have further conducted related calculations for the HOMO and LUMO distributions of $\text{Cu}_3\text{-BTDE-COF}$ (Figure 4c, Figure S32 and S33). For $\text{Cu}_3\text{-BTDE-COF}$, its HOMO is mainly distributed on BTDE, while LUMO is mainly distributed on Cu_3 (Figure 4c). This also indicates that Cu_3 has a stronger affinity for electrons when compared with BTDE and Cu_3 and BTDE are the active centers of CRR and CER, respectively.

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Afterwards, we have also conducted the calculations of density of states to demonstrate the relationship between CO₂ and Cu₃-BTDE-COF (Figure 4d and Figure S34). The calculation results show that the energy level of CO₂ is located within the band gap of Cu₃-BTDE-COF, thus there is a high possibility of electron transfer to CO₂. At the same time, it also verifies the good coupling between Cu₃-BTDE-COF and CO₂, which is much beneficial for CRR (Figure 4d)^[6]. Then, theoretical calculations have been further performed to evaluate the possible interaction or diffusion effect of Li⁺ (Figure 4e). Benefiting from the abundant functional groups in Cu₃-BTDE-COF, they would serve as hopping sites to be interacted with Li⁺. We have calculated the possible adsorption site of Li⁺ in Cu₃-BTDE-COF and the results show that Li⁺ is more easily adsorbed near cyano group (Figure 4f). Beyond that, the Li⁺ diffusion paths in the pore channels have also been calculated, and it tends to diffuse to the site of cyano group, supporting above-mentioned results (Figure 4f). Therefore, these results demonstrate the synergistic roles of Cu₃, BTDE and cyano group in promoting the electron-hole utilization, CO₂ adsorption/activation and Li⁺ interaction/diffusion for Cu₃-BTDE-COF, which is much beneficial for the CRR and CER to largely improve the battery performance.

Conclusions

In summary, a series of MCOFs (i.e. Cu₃-BTDE-COF, Cu₃-BTDA-COF and Cu₃-DT-COF) with multiple active sites are synthesized by in-situ covalent connection of Cu₃ and BTDE, which can be utilized as efficient photocathode catalyst for light-assisted Li-CO₂ battery. Among them, Cu₃-BTDE-COF is a newly designed sp²-carbon-linked based MCOF covalently linked by Cu₃ and thiazole-cyano group (BTDE), in which the existence of sp²-carbon bond improves the stability of the skeleton and prolongs the π - electron conjugation time on the skeleton plane, meanwhile multiple active units (i.e., Cu cluster, thiazole and cyano sites) are beneficial to the CO₂ activation and evaluation processes. Specifically, Cu₃-BTDE-COF based battery delivers an ultra-low overpotential (0.18 V) and excellent round-trip efficiency (95.2%), outperforming the Schiff base COFs (i.e. Cu₃-BTDA-COF and Cu₃-DT-COF) and majority of the reported photocathode catalysts. Furthermore, theoretical calculations and characterization have shown that Cu₃-BTDE-COF with metal active centers, thiazole and cyano multifunctional groups possess strong CO₂ adsorption/activation and Li⁺ interaction/diffusion ability to boost the CRR/CER kinetics and related battery efficiency. This work demonstrates the importance of sp²-carbon-linked based MCOF by comparing the properties of Cu₃-BTDE-COF, Cu₃-BTDA-COF and Cu₃-DT-COF applied to light-assisted Li-CO₂ battery. It provides a reference for the application of different bonded COFs in light-assisted battery techniques.

Supplemental Information includes 34 figures, 5 tables.

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Conflict of Interest

The authors declare no conflict of interest.

Keywords: Light-assisted CO₂ energy storage • Redox molecular junction • Photocatalysis • Metal-covalent organic frameworks

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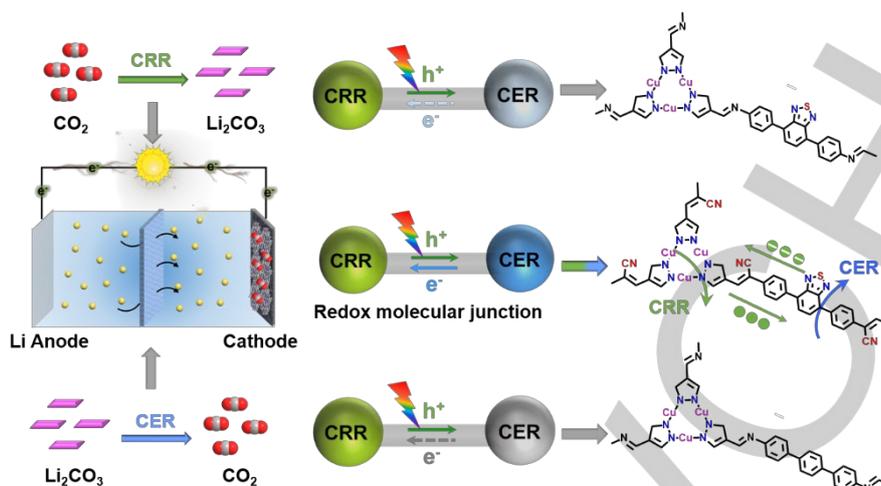
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A kind of redox molecular junction sp^2 carbon-conjugated metal-covalent organic framework with multiple active sites have been prepared and can be successfully applied for light-assisted CO₂ energy storage.